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# Lanthanide polysulfides at high pressures

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## **Abstract**

High pressure behavior of selected stoichiometric lanthanide disulfides and sesquisulfides is studied in a diamond cell by X-ray diffraction, Raman spectroscopy, and optical reflectivity. Upon compression to 20 GPa, there occurs a reversible phase transition in layered  $\alpha'$ -LnS<sub>2</sub> (Ln: La–Nd) rare earth disulfides (P2<sub>1</sub>/b, C<sub>2n</sub>, Z=4) with the distorted anti-Fe<sub>2</sub>As structure (the PbFCl type, P4/nmm,  $D_{4h}^7$ , Z=2) to higher symmetry superstructures of the parent anti-Fe<sub>2</sub>As and  $\alpha'$ -PrS<sub>2</sub>, and at about 8 GPa for  $\alpha'$ -NdS<sub>2</sub> is associated with changes in the scheme of direct optical transitions.  $\gamma$ -Gd<sub>2</sub>S<sub>3</sub> with the defect Th<sub>3</sub>P<sub>4</sub> structure ( $\overline{A}3d$ ,  $Z=4$ ) remains stable to <u>at</u> l of intermediate phases of the  $Tm_2S_3$ -II ( $Lu_2S_3$ -II,  $P_2/m$ ),  $U_2S_3$  ( $Lu_2S_3$ -III,  $Pnma$ ), and NdYbS<sub>3</sub> ( $Lu_2S_3$ -IV,  $Pnma$ ) types up to 20 GPa. All the compounds remain semiconducting in the entire pressure range covered in this study.  $\oslash$  2001 Elsevier Science B.V. All rights reserved.

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compounds and wide-gap high-resistivity semiconductors. formation on the phase stability and occurrence of fully They have potential application as far-infrared transmitting reversible phase transformations. materials, lasers, phosphors, scintillators, and magnetooptical devices [1–3]. These compounds are also precursors for several crystalline and amorphous systems. Much effort has been concentrated on techniques to **2. Phase transformations in lanthanide disulfides:** process them and to tune their optical and electronic **Raman studies** properties.

In this study, the effect of pressure on the structures and<br>optical properties of lanthanide disulfides and sesquisul-<br>fides is examined for selected compounds using X-ray<br>anti-Fe<sub>2</sub>As type  $(P4/nmm, D_{4h}^T, Z=2, a_0 \approx 4 \text{ Å}, c$ diffraction, Raman spectroscopy, and optical reflectivity In this structure, layers of five chalcogen atoms in the basal measurements in a diamond anvil cell. More details on the face-centered square plane (the Flahaut plane) are sepa-<br>structural transitions, along with the description of sample rated by two slabs of alternating  $\text{Ln}^{3+}$ elsewhere  $[4-6]$ . The optical reflectivity spectra presented distorted from this due to the formation of X–X pairs show a correlation between structural and electronic within the basal planes [7]. This distortion gives rise to a properties as these compounds undergo pressure-induced number of pseudo-cubic, tetragonal, orthorhombic, and phase transitions. Several quenchable polymorphic modi-<br>fications of these compounds, depending on temperature covalent binding in the S-S dimers accounts for semiconand pressure conditions of synthesis, have already been ducting properties of these materials.

**1. Introduction** reported. However, these studies do not give any direct (in situ) evidence about the actual transformations and phase Rare-earth disulfides and sesquisulfides are refractory equilibria. Diamond cell studies provide invaluable in-

covalent binding in the S–S dimers accounts for semicon-

Different polymorphs of stoichiometric light rare earth <sup>\*</sup>Tel.: +49-11-711-689-1404; fax: +49-11-711-689-1444. disulfides (La–Nd) are obtained at ambient pressure but *E-mail address:* andrzej@servix.mpi-stuttgart.mpg.de (A. Grzechnik). variable temperatures [8–11]: α (α') (

(*Pnma, D*<sup>16</sup><sub>2h</sub>, Z=8),  $\gamma$  (*P4*/*nmm*, *D*<sup>7</sup><sub>4h</sub>, Z=2). Preparation  $\alpha'$ -NdS<sub>2</sub>. In addition to this, in every case a new band of stoichiometric heavier rare earth disulfides requires high appears at about 430 cm<sup>-</sup> tetragonal products are recovered to ambient conditions. at each case.

at and near ambient conditions are similar, the only the pseudo-cubic and tetragonal polymorphs at ambient differences being in the absolute positions of the respective conditions. The experimental observations presented here bands (the selection rules:  $\Gamma = 9A_g + 9B_g + 8A_u + 7B_u$ , can be interpreted as an evidence for sluggish pressurewhere the A<sub>g</sub> and B<sub>g</sub> modes are Raman active, and the A<sub>u</sub> induced phase transitions in layered  $\alpha'$ -LnS<sub>2</sub> (Ln: La–Nd) and B<sub>u</sub> modes are infrared active) [4]. Fig. 1 shows rare earth disulfides ( $P2_1/b$ ,  $C_{2h}^5$ , strongest bands at about 400–430 cm<sup>-1</sup> are well resolved ascence of symmetry for the anti-Fe<sub>2</sub>As superstructure.<br>at lowest pressures for all the compounds. Their scattering The mechanism for this structural change can b intensity appears predominantly for parallel polarization visioned as a tendency of the 9-fold polyhedra around the vectors of incident and scattered photons ( $E_i$  II  $E_s$ ) [13].  $Ln^{3+}$  ions to become more symmetric with  $\text{Ln}^{3+} \text{S}^{2-} (\text{S}_2^{2-})_{1/2}$ . Hence, the strong Raman bands are<br>attributed to the S-S stretching vibration of the  $(\text{S-S})^{2-}$  the observed Raman bands at the respective pressures, it ions ( $A_g + B_g$ ). Upon compression, these two bands merge could be argued that the number of the molecular units and the higher frequency components have negative pres-<br>does not change ( $Z=4$ ) and the resulting high pressur sure shifts. This band convergence occurs at about 5 GPa structure is a pseudo-cubic or tetragonal anti-Fe<sub>2</sub>As superfor  $\alpha'$ -LaS<sub>2</sub>,  $\alpha'$ -CeS<sub>2</sub>, and  $\alpha'$ -PrS<sub>2</sub>, and at about 8 GPa for structure.

pressure techniques [12]. Depending on the conditions of decompression to ambient conditions, all the spectral the reaction of the respective elements, pseudo-cubic or changes are reversible with a hysteresis of about 3–4 GPa

The actual pressure necessary to synthesize stoichiometric Analysis of the Raman spectra of the  $\text{LnS}_2$  compounds compounds increases along the Sm-Lu rare earth series. with different superstructures [11] reveals the absence of<br>The first-order Raman spectra of  $\alpha'$ -LnS, (Ln: La–Nd) splitting of the strongest band at about 400–430 cm The mechanism for this structural change can be endoes not change  $(Z=4)$  and the resulting high pressure



Fig. 1. Raman spectra upon compression (a) and pressure dependence of the observed Raman frequencies (b) in NdS<sub>2</sub>. All spectra are vertically offset for clarity.

Dy have three polymorphs at atmospheric pressure [14,15]. The high temperature  $\gamma$  phases have a lattice of the Th<sub>3</sub>P<sub>4</sub> All the previous reports on the high pressure synthesis type (*IA3d*, *Z*=4), with 8-fold coordinated cations random- of the lanthanide sesquisulfides have ly distributed over available sites,  $\text{Ln}_{2.67 \otimes 0.33}$   $\text{S}_4$  ( $\otimes_{0.33}$  is covered samples, i.e. metastable compounds at atmos-<br>the number of vacancies). Heavier rare earth sesquisulfides pheric conditions [14,15] the number of vacancies). Heavier rare earth sesquisulfides  $(Ln=Ho-Lu)$  have six different structural types, depend- not provide any direct (in situ) evidence about the actual ing on cation radii and coordination numbers, and tempera- transformations and phase equilibria. The ture and pressure conditions [14,15]. In these materials, the corundum→thorium phosphide transition for Ln<sub>2</sub>S<sub>3</sub> is coordination numbers of the cations vary from six to eight postulated based on the structural considerations [5,6]. and the higher coordinated polymorphs are obtained at  $\gamma$ -Ln<sub>2</sub>S<sub>3</sub> compounds have the smallest reduced volumes high pressures. At ambient conditions,  $Lu_2S_3$  takes the and the highest coordination of the cations at ambient corundum structure ( $R\overline{3}c$ ) with 6-fold coordinated  $Lu^{3+}$  conditions, so that all the other polymorphs cations ( $\varepsilon$ -Lu<sub>2</sub>S<sub>3</sub> or Lu<sub>2</sub>S<sub>3</sub>-I). Other polymorphs with this transform to the Th<sub>3</sub>P<sub>4</sub> structure at high pressures. composition are synthesized at high pressures and high Fig. 2 displays diffraction patterns of  $Lu_2S_3$  up to 23.0 temperatures [14,15]. The  $Lu_2S_3$ -II phase of the  $Tm_2S_3$ -II GPa at room temperature. The diffractogram temperatures [14,15]. The  $Lu_2S_3$ -II phase of the  $Tm_2S_3$ -II type  $(P2_1/m$ , coordination numbers of the cations are 8, 7,<br>above about 5 GPa can be indexed with the cubic cell of<br>and 6) is obtained at about 1.0–3.0 GPa and 1273 K. The<br>the Th<sub>3</sub>P<sub>4</sub> type  $(I\overline{4}3d)$  [5]. There is no Lu<sub>2</sub>S<sub>3</sub>-III modification with the U<sub>2</sub>S<sub>3</sub> structure (*Pnma*,

**3. Phase transformations in lanthanide sesquisulfides:** coordination numbers  $7$  and  $7+1$ ) is recovered from **X-ray diffraction studies** somewhat higher pressures, but approximately the same temperatures. The  $Lu_2S_3$ -IV variant of the NdYbS<sub>3</sub> type is Rare earth sesquisulfides  $Ln_2S_3$  in the series  $Ln=La-$  found at about 3.0 GPa and at temperatures above 1573 K v have three polymorphs at atmospheric pressure [14,15]. (*Pnma, coordination numbers 8 and 6*).



Fig. 2. Energy-dispersive X-ray diffraction patterns of Lu<sub>2</sub>S<sub>3</sub> (a) and  $\gamma$ -Gd<sub>2</sub>S<sub>3</sub> (b) as a function of pressure ( $E^*d$ =70.584±0.072 keV Å). Miller indices are given for the Th<sub>3</sub>P<sub>4</sub>-type phase of Lu<sub>2</sub>S<sub>3</sub> at 8.3 GPa ( $\gamma$ -Lu<sub>2</sub>S<sub>3</sub>: *IA3d, a* = 7.9957±0.0026 Å, *V*=511.18±0.50 Å<sup>3</sup>). All patterns are vertically offset for clarity.

between the corundum and  $\text{Th}_3\text{P}_4$  types. The data of this **4. Optical properties** study show that  $\varepsilon$ -Lu<sub>2</sub>S<sub>3</sub> with the corundum structure at atmospheric conditions (Lu<sub>2</sub>S<sub>3</sub>-I) does undergo a phase and NdYbS<sub>3</sub> (*Pnma*) types does require high temperatures

structural and electronic transformations in  $\gamma$  group of rare gaps in  $\alpha$ -LnS<sub>2</sub> are about 2–2.5 eV for direct transitions earth sesquisulfides is still open.  $\gamma$ -Gd<sub>2</sub>S<sub>3</sub> occurs towards and 2 eV for indirect transit earth sesquisulfides is still open.  $\gamma$ -Gd<sub>2</sub>S<sub>3</sub> occurs towards. the end of the series of the Th<sub>3</sub>P<sub>4</sub>-structured rare earth Fig. 3 shows optical reflectivity spectra of NdS<sub>2</sub> upon sesquisulfides that can be obtained using simple synthesis compression. The red shift of the high energy reflectivity methods [16] and does not require high pressure [14,15] or edge indicates that the lowest energy direct interband attrition milling [17] techniques of preparation. As such, it transitions shift towards lower energies. The reflectivity in could serve as a model compound for the entire lanthanide the low energy infrared region does not increase due to series at high pressure conditions. Combined X-ray diffrac- free carriers, as a result of a transition (or a tendency tion (Fig. 2) and Raman spectroscopy data [6] provide the towards it) into a metallic state. The reflectance spectra evidence that there is no structural transition in  $\gamma$ -Gd<sub>2</sub>S<sub>3</sub> suggest that the oscillator strength just above the band gap upon compression to 30 GPa. One of the mechanisms for a due to the direct transitions decreases significantly with possible structural change is an increase of coordination increasing pressure up to about 9 GPa. At higher pressures, around the gadolinium atoms. This would be a driving the optical edge broadens and the reflectivity at high force to achieve a higher density phase. The stability of energies rises. This would be related to a new scheme of  $\gamma$ -Gd<sub>2</sub>S<sub>3</sub> at high pressures (at least to 30 GPa) suggests direct optical transitions at the higher symmetry anti-Fe<sub>2</sub>As that it would not be the case for the entire sesquisulfide superstructure. group and that the 8-fold cation coordination is the The fundamental optical band gaps in the  $\gamma$ -Ln<sub>2</sub>S<sub>3</sub> compounds originate from the separation between the

The calculations of the electronic structure in  $\text{LnS}_2$  with transition into the Th<sub>3</sub>P<sub>4</sub> structure above 5.3 GPa (Lu<sub>2</sub>S<sub>3</sub>-V the linear muffin-tin orbital method (LMTO) [18] reveal or  $\gamma$ -Lu<sub>2</sub>S<sub>3</sub>). This fully reversible transition ( $\varepsilon \leftrightarrow \gamma$ ) occurs that the states near the bottom of the conduction band are at room temperature and does not involve any inter- formed mainly by the 5*d* orbitals of the lanthanum atoms, mediates with cationic coordination numbers between six but there are also considerable (up to 30%) admixtures of and eight  $(Lu_2 S_3$ -II,  $Lu_2 S_3$ -III,  $Lu_2 S_3$ -IV). This suggests the 4*f* orbitals. The top of the valence band is formed by that the occurrence of the Tm<sub>3</sub>S<sub>3</sub>-II ( $P_2/m$ ),  $U_2 S_3$  ( $Pnma$ ), antibonding (relative to th that the occurrence of the Tm<sub>2</sub>S<sub>3</sub>-II ( $P_2/m$ ),  $U_2S_3$  (*Pnma*), antibonding (relative to the S–S interaction) combinations and NdYbS<sub>3</sub> (*Pnma*) types does require high temperatures of the  $\pi$  bonds (parallel to the at high pressures. dimers. The calculated band gap of 0.95 eV corresponds to The question of stability and further pressure-induced indirect transitions  $\Gamma$ –R, R–X, or M– $\Gamma$ . The measured

compounds originate from the separation between the

 $0.05$ 20.4 GPa  $0.040$ 2.8 GPa  $Lu<sub>2</sub>S<sub>3</sub>$ Nd<sub>S2</sub> 6.2 GPa 9.7 GPa 0.035  $0.04$ 6.6 GPa 20.6 GPa 0.030 18.7 GPa 3.5 GPa 16.2 GPa Reflectivity (R<sub>d</sub>) Reflectivity (R<sub>d</sub>)  $0.025$  $0.03$ 12.6 GPa 4.4 GPa 9.2 GPa 0.020 5.4 GPa  $0.02$ 0.015 7.4 GPa 0.010  $0.01$ 0.005  $0.000$  $0.00C$  $\overline{4}$  $\mathbf 0$  $\overline{c}$  $\overline{3}$  $\overline{4}$ Energy (eV) Energy (eV)

0.045

Fig. 3. Optical reflectivity spectra of NdS<sub>2</sub> and Lu<sub>2</sub>S<sub>3</sub> at high pressures. The reflectivity  $R_d$  is measured at the diamond–sample interface. There is no vertical offset of the spectra.

3*p*(S) valence band at the M point and the 5*d*(Ln) **Acknowledgements** conduction band at the  $\Gamma$  point of the Brillouin zone [6]. Optical reflectivity spectra in  $\gamma$ -Gd<sub>2</sub>S<sub>3</sub> up to 20 GPa show Thanks to K. Syassen for his support and several a red shift of the lowest energy direct interband transitions. discussions and U. Oelke for his assistance with optical At higher pressures, broadening of the reflectivity spectral reflectivity measurements. Part of the work presented here features is associated with a loss of Raman intensity as a was carried out in Ecole Normale Superieure de Lyon and ´ result of a possible transformation in the electronic struc- Laboratoire de Physico-Chimie des Materiaux Lumines- ´ ture. The optical reflectivity spectra in  $Lu_2S_3$  at high cents, Universite Claude Bernard Lyon I. pressures are shown in Fig. 3. At atmospheric pressure, the optical absorption edge for  $\varepsilon$ -Lu<sub>2</sub>S<sub>3</sub> with a direct band gap is 3.18 eV [19]. The blue-to-red change in the shift of the **References** high energy reflectivity edge is well correlated with the corundum-thorium phosphide structural phase transition [1] P.N. Kumta, S.H. Risbud, J. Mater. Sci. 29 (1994) 1135. and corresponds to a major change in the electronic [2] C. Witz, D. Huguenin, J. Lafait, S. Dupont, M.L. Theye, J. Appl. There is no contracture in this material Similarly to a Gd S. there is no Phys. 79 (1996) 2038. structure in this material. Similarly to  $\gamma$ -Gd<sub>2</sub>S<sub>3</sub>, there is no<br>evidence for the increase of reflectivity in the low energy [3] R. Mauricot, J. Bullot, J. Wery, M. Evain, Mater. Res. Bull. 31<br>(1996) 263. infrared region in  $Lu_2S_3$  due to an emergence of free [4] A. Grzechnik, Physica B 262 (1999) 426. carriers. [5] A. Grzechnik, J. Alloys Comp. 299 (2000) 137.

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